

Comment on “Electronic Structure and Optical Properties of Si-Ge Superlattices”

We have self-consistently calculated¹ the electronic structure of the strained Si₄Ge₄/Si(001) superlattice, interpreting the electroreflectance transitions² *A* (observed 1.24 eV, calculated 1.26 eV) and *B* (observed 1.8 eV, calculated 1.74 eV) as folded-in (pseudo) direct excitations [Figs. 1(a)–1(c)], in accord with Ref. 2. In contrast, transition *I*, observed at 0.8 eV and interpreted previously² as the *direct* band gap, was shown to be an indirect $\bar{\Gamma}_v \rightarrow \bar{\Delta}_c$ excitation either to the superlattice Δ_c state (calculated¹ at 0.92 eV) or to the substrate Δ_c state (calculated at 0.80 eV). Other calculations³ and experiments⁴ support this assignment. More recently, Wong *et al.*⁵ carried out non-self-consistent calculations within the empirical pseudopotential method. The central point of their Letter was to note that if larger Ge-Ge interplanar spacings were postulated (making the Ge–Ge bond lengths longer and more bulklike), the direct superlattice band gap (transition *A*) was reduced from 1.2 to 0.9 eV, close to the value where transition *I* was seen.² On the basis of this empirical adjustment, they proposed that the observed³ transition *I* is in fact the *direct* band gap (*A*), in conflict with more recent experiments.⁴ This parameter adjustment was not constrained by any (variational or microscopic) principle; the only motivation was that it might mimic effects of defects at Si/Ge interfaces. However, such interfacial defects are more likely to destroy coherence with the substrate, removing the biaxial epitaxial constraint ($a_{\parallel} = a_{\text{Si}}$), than to permit uniaxial relaxation of Ge–Ge bonds. In both cases Ge–Ge bonds relax, but in the former the Ge-Ge interplanar distance naturally becomes *smaller*. We show below that (i) even if the assumed *increase* in Ge-Ge interplanar distances were correct, the results of Wong *et al.* lead to further inconsistencies with the data, and (ii) to make the system more nearly direct, one needs to *reduce* the Ge-Ge interplanar spacing.

The observations pertinent to the understanding of the direct and indirect gaps in this system are the following: (i) Strain splits the conduction band at *X* into X_c^{\perp} and X_c^{\parallel} ; the valence-band maximum at Γ splits into Γ_v^{\perp} and Γ_v^{\parallel} (here \perp and \parallel denote directions perpendicular and parallel to the interface, respectively). (ii) For compressive strain [a Si substrate, Figs. 1(a) and 1(b)], X_c^{\parallel} is below X_c^{\perp} and Γ_v^{\perp} is below Γ_v^{\parallel} , but for tensile strain [a Ge substrate, Figs. 1(e) and 1(f)], X_c^{\perp} is below X_c^{\parallel} and Γ_v^{\parallel} is below Γ_v^{\perp} . (iii) By symmetry, only X^{\perp} folds into $\bar{\Gamma}$ for [001]-oriented superlattices.

When the Ge-Ge interplanar spacing is increased (by assumption⁵), the conduction-band ($X_c^{\perp} - X_c^{\parallel}$) and the valence-band-maximum ($\Gamma_v^{\parallel} - \Gamma_v^{\perp}$) splittings increase too. In the calculation of Wong *et al.*, Γ_v^{\parallel} was raised by 0.3 eV more than X_c^{\perp} , reducing by this amount the energy of transition *A*. However, the same effect must also

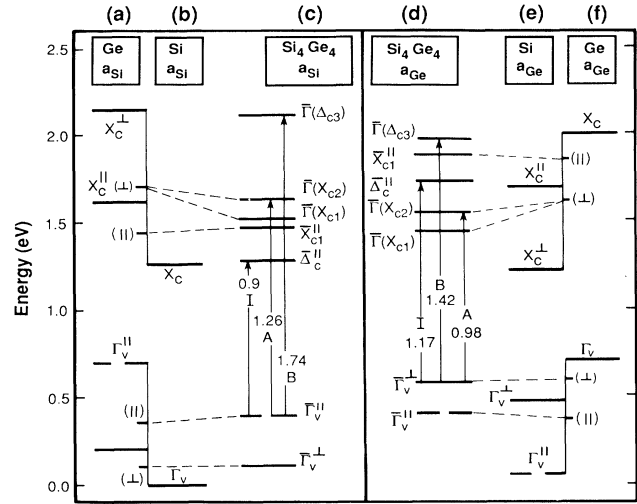


FIG. 1. Electronic energy levels for Si₄Ge₄ superlattice and its constituents on Si and Ge substrates. Band offsets are from Ref. 1.

reduce the energies of I ($\Gamma_v^{\parallel} - \bar{\Delta}_c^{\text{min}}$) and *B* [$\Gamma_v^{\parallel} - \bar{\Gamma}(\Delta_{c3})$] to below 0.8 eV and⁵ to 1.4 eV, respectively. This creates a substantial conflict with the recently observed indirect transition (*I*) at 0.8 eV,⁴ and fails to account for the pseudodirect transition (*B*) at⁴ 1.8 eV. These are naturally interpreted within our model as transition *I* and *B*, respectively (calculated at ~ 0.8 and 1.74).

Our first-principles calculation [Figs. 1(d)–1(f)] shows that to make the system more nearly direct one needs to invert the order of X_c^{\perp} and X_c^{\parallel} and that this can be achieved by *reducing* the Ge-Ge interplanar spacing, *not* by increasing it.⁵ The former condition is achieved on a Ge substrate [Fig. 1(d)], for which our calculation shows $I > A$. A similar trend is expected on a Si substrate when defects destroy coherence with the substrate.

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