

Mills Boulder II

I-8

©mills 2008

Discussion so far: - surfaces (all controlled)

QW: How "passivate" a TMO surface
- speculation

Rest of lectures: super-lattice. Apparently better controlled system

Ohtomo, Muller-Grazul Hwang. Nature 419 378 (2002)
"Oxide epitaxy" = "careful pulsed laser deposition"

Idea $\cup \cup \cup$

— substrate

Shoot calibrated "puffs" of ions at substrate, under appropriate oxygen pressure. Monitor result by high class scattering

- Can grow many "materials by design"

So far: mainly studied variants on " ABO_3 " perovskite structure.

- B site: simple cubic lattice. Latt. par $\approx 4\text{\AA}$
- O: in between each pair of B's
- A: body center of B cube

Typically: B site: electronically active
A site: controls carrier conc. on B

Simplest superlattice: charge only A

001 Superlattice: pick 1 cubic axis. Alternate A site:

Ohtomo et. al. $(\text{SrTiO}_3)_m / (\text{LaTiO}_3)_n$
 Band insulator: $\text{Ti } d^0$ "Mott" insulator: $\text{Ti } d^1$

Advantageous structures: need only keep La/Sr in place.

TEM image: Thinned sample. Avg over $\sim 1000 \text{ \AA}$

- Interface appears v. flat

Qn: How much local disorder (Sr/La exch) other

- Qn Charge: Sr Sr La La La Sr Sr

Ti Ti Ti Ti Ti Ti
 0 .5 1 1 .5 0

each if 0 all stoichiometry good

each La $\Rightarrow 1 e^- \Rightarrow 3 e^-$.

where sit.? Natural Guess: edge

of La region: $\approx .5 e^-$

Obvious questions:

how wide is edge

how thick must you make central region to get γ

what can you get edge to do?

TEM-EELS: Inelastic spectrum corr. to
transitions from Ti L edge to unocc Ti d states
(?)

~~Id~~

($\Rightarrow$) inverse photoemission + excitonic correct
Cint. bet. ^{core} hole & el. in unocc. states

Idea: diff Ti valence $\Rightarrow$ diff. occ Ti d-states
 $\Rightarrow$ diff final state energies

($\Rightarrow$) differences in spectra large of electronic
hopping time. Will see explicit ex later)

Found (for these lattices): $n \text{Ti}^{3+} = n \text{La}$

Also: mecs Rxy: $2/3 e^-/\text{La}$. Qn: some
carrier trapped
? Compensation

($\pi - 1$)

Polarization "Catastrophe"

Simple observation (Sawatzky): most TMO planes are polar = charged

ex) 100 LaTiO_3 . Formal valence $\text{La}^{3+} \text{Ti}^{3+} \text{O}^{2-}$
 Planes $\text{TiO}_2^{3+ \ 2 \times 2-} = -1$ $\text{LaO}^{3+ \ 2-} = +1$

La $\# \text{La} \neq \# \text{Ti}$ Net charge $\Rightarrow$ Very Bad
 Ti

La But even if $\# \text{La} = \# \text{Ti}$ - problem
 Ti

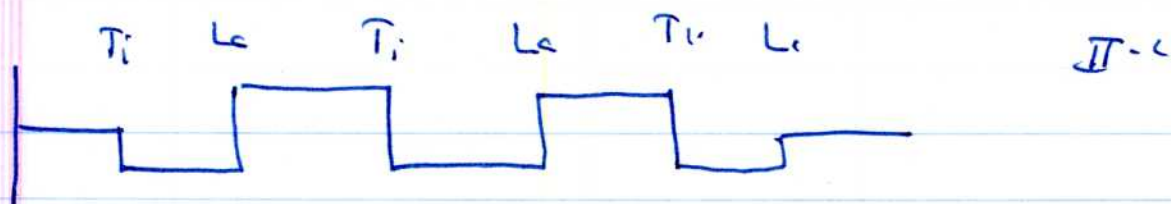
La Poisson: $\nabla \cdot \mathbf{E} = \frac{4\pi\rho}{\epsilon}$
 Ti

2d charged sheet: $\rho = \sigma \delta(z - z_n)$
 $\Rightarrow$ across sheet $\Delta E = \frac{4\pi\sigma}{\epsilon}$



$\Rightarrow N$ Layers, L.C.C
 $\Delta V = \frac{N 4\pi e^2 \sigma c}{\epsilon z}$

Typical resolution: surf. layer off
 stoichiometry. $\text{TiO}_2 \rightarrow \text{TiO}_{1.85} \Rightarrow -1/2$
 $\text{LaO} \rightarrow \text{La}_{1.83}\text{O} \Rightarrow +1/2$



Principal reason why TMO surfaces are typically bad

Sawatzky realized: can compensate polarization by moving electrons instead

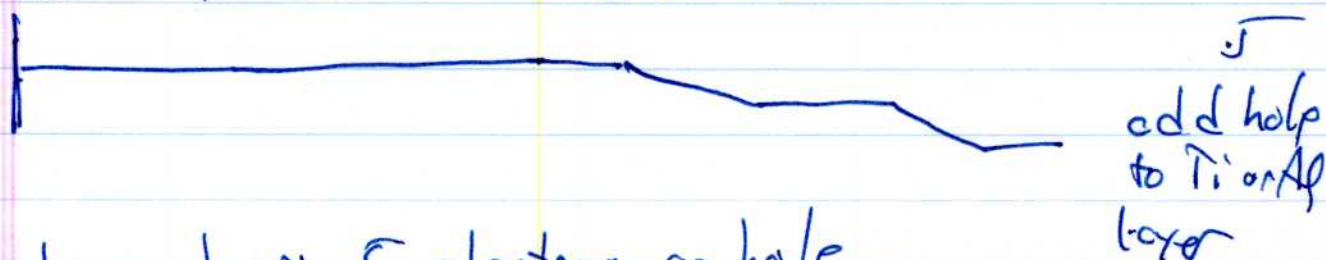
Consider structure: $(\text{LaAlO}_3)_n (\text{SrTiO}_3)_m$

$\left[\begin{array}{l} \text{LaAlO}_3: \text{ wide bandgap insulator} \\ \text{effective barrier to electrons} \\ \text{SrTiO}_3: \text{ very small gap to add el to} \\ \text{Ti d-states (under gap to add holes)} \\ \text{NON Polar. } \text{Sr}^{2+} \text{Ti}^{4+} (\text{SrO})(\text{TiO}_2)_{\text{net}} \end{array} \right.$
 Materials v. common in technology

Sr Ti Sr Ti La Al La Al ...



Sr Ti Sr Ti Sr Al La Al



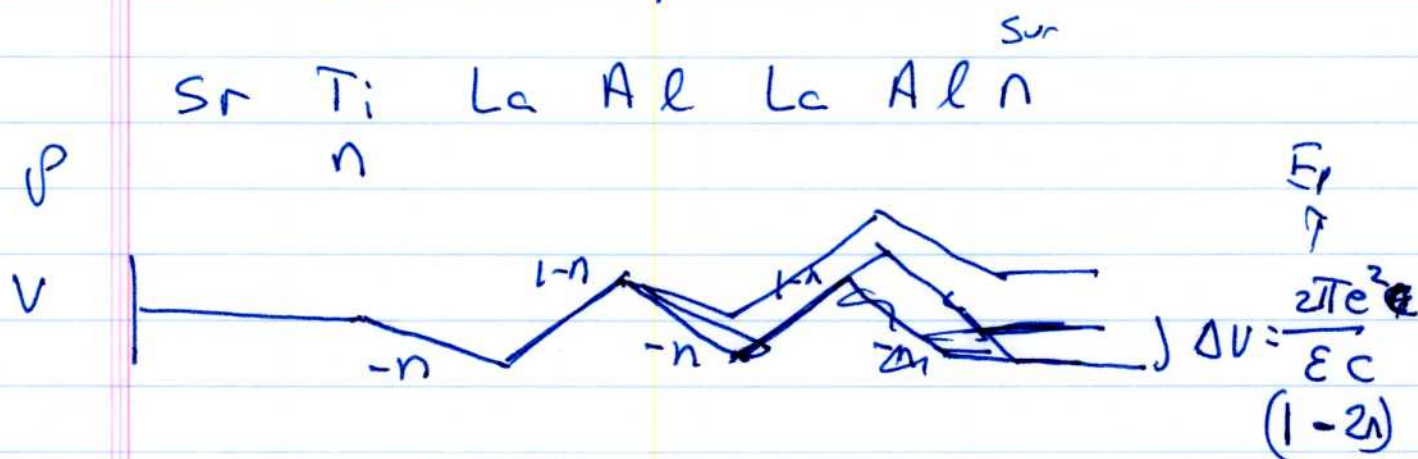
$\Rightarrow$ doping by $\approx .5$ electron or hole from pol. catalysts

Examine electrostatics in more detail.
Simple model.

Energy to put n electrons on STO in absence of field

$$E = E_D n + \frac{1}{2} E_C n^2$$

if have n el., must have surf. charge layer
and also have dipole field



⇒ Dipole. Energy ~ Dipole squared ~ E_D

$$E = E_D n + \frac{1}{2} E_C n^2 - N E_p (1-2n) n + \frac{1}{2} N^2 E_D (1-2n)^2$$

Minimize : $E_D + E_C n - N E_p - 2 N^2 E_D n + 4 N^2 E_D n^2 + 2 N E_p n$

$$\left(\frac{N E_p + 2 N^2 E_D - \Delta}{E_C + 4 N E_p + 4 N^2 E_D} \right) = n / \text{cell}$$

Characteristics: Threshold.
Saturation at $1/2$ Approach rapid if $E_D > E_r$

Examine charge dist. in more detail

Classical appx:

Charge density n_j

~~Ex~~

$$E = \sum_j \Delta n_j + \frac{1}{2} E_c n_j^2 + 2E_p \sum_{j=0}^{\infty} \sum_{l=j+1}^{\infty} n_j n_l |j-l|$$

• Put all charge in layer 0

$$E = \Delta n + \frac{1}{2} E_c n^2$$

Put δ in layer 1

$$\begin{aligned} E &= \Delta n + \frac{1}{2} E_c (n - \delta)^2 + \delta^2 + E_p (n - \delta) \delta \\ &= \Delta n + \frac{1}{2} E_c n^2 - E_c n \delta + E_c \delta^2 + E_p n \delta - E_p \delta^2 \end{aligned}$$

$\Rightarrow$ only if $E_c > E_p$ does it pay

$$\begin{aligned} &\sum_l l a r^l \\ &\quad \quad \quad \frac{d}{dr} \sum_l r^l \\ &= \frac{r}{(1-r)^2} \end{aligned}$$