

Surfaces & Interfaces (II) [Millis]

- We shall consider explicitly superlattice of perovskite ABO_3

▲ perovskite { A sits on corner of cubic lattice
 { B sits on center of cubic lattice
 { O sits on the faces of cubic lattice

▲ We shall assume B transition metal, & A = La, Sr, ...

▲ The general superlattice is 001 (along cubic axis)

- Homometallic: "A" changes, "B" stays the same

e.g. $LaTiO_3 / SrTiO_3$

- Heterometallic: "B" changes ("A" may or may not change)

case A: hard wall, e.g.

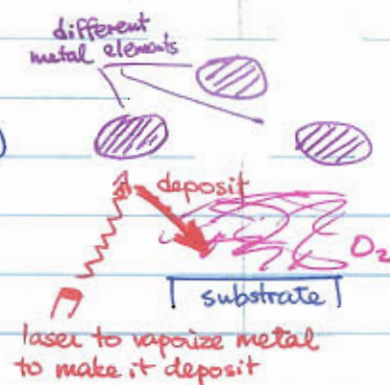
case B: different correlated material, e.g. $LaMnO_3 / YBa_2Cu_3O_7$

- Fabrication methods (all variation on film)

▲ Oxide epitaxy

▲ MBE

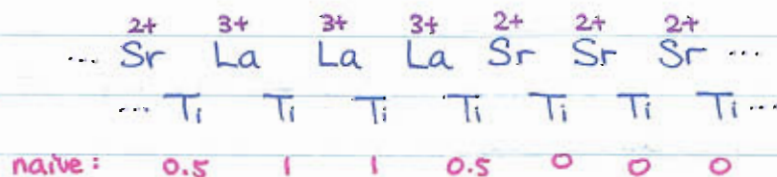
▲ Pulsed Laser Deposition



- Image can be obtained because different atoms have different cross sections.

▲ But reality is often less "clean & clean" than the image suggests.

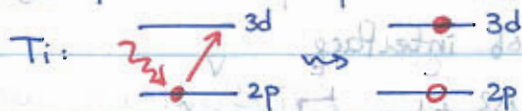
- Consider



▲ $LaTiO_3$ is Mott insulator, $SrTiO_3$ is band insulator

▲ The charge distribution can be measured by TEM/EELS

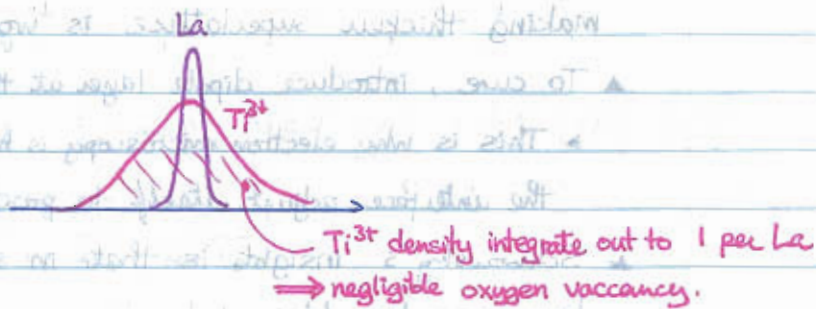
► look for inelastic processes, in particular:



probability depends on what's already in d-shell.

► The Ti state with (say) $+0.5$ charge can be decompose into superposition of $+0$ & $+1$ charge (not clear why it is just superposition)

► Result:



▲ One can also measure longitudinal Hall resistance to find carrier density.

► Get $\sim 2/3$ carrier per La naively. Still close to ~ 1 as expected.

▲ Now it is also possible to do optical measurement (drude peak)

▲ In above TEM can truly resolve individual layer, while Hall probe/optical measure involve averaging between layers.

• Magnetization can be seen at interface

• Consider $\text{LaMnO}_3 / \text{YBa}_2\text{Cu}_3\text{O}_7$

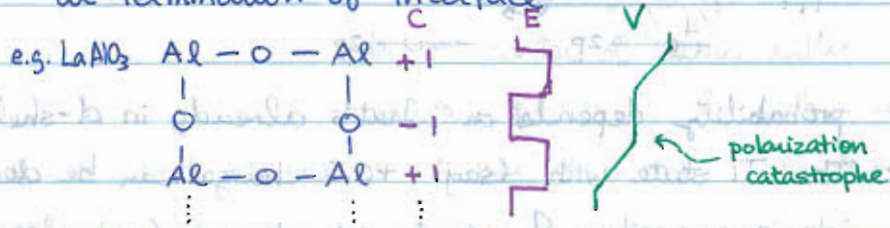
▲ Orbital occupation can be measured using polarized light.

▲ Bulk: holes found in $|x^2-y^2\rangle$ state

Interface: holes found ALSO in $|3z^2-r^2\rangle$ state

• Consider electrostatics of these superlattice

▲ Most ABO_3 perovskite are polar $\Rightarrow$ charges build up at termination of interface



▲ Intuitively, this is why from material science perspective, for thicker polar layer in superlattice, quality of interface reduces.

▲ To cure, introduce dipole layer at the end: $C: \frac{1}{2} -1 +1 \dots -1 +1 -\frac{1}{2}$

► This is why electron microscopy is hard to do in these material, since the interface adjust itself to produce the desired polar change.

▲ Sawatzky insight is that in superlattice, the $\pm 1/2$ charge layer can be obtained by having an 2D e^- gas in the interface (hence interface $\Rightarrow$ more metallic).

▲ Question: Do we have ion transfer or e^- transfer?

▲ For $SrTiO_3/LaAlO_3$, we have p-type & n-type

p-type: $\dots Ti^{+1/2} Sr^{+1} Al^{-1} La \dots$

n-type: $\dots Sr^{-1/2} Ti^{+1} La^{-1} Al \dots$

▲ For n-type find $-0.6 e^-$ & $\sim 0.15 O$ vacancy

For p-type find $0 e^-$ & $\sim 0.3 O$ vacancy

$\Rightarrow$ for some reason the material does not want to dope interface with holes.

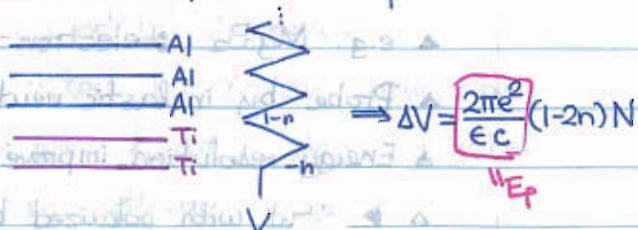
• Heterometallic interface is less sharp than homometallic

• For $LaAlO_3/SrTiO_3$, the interface goes superconducting, but mobile carrier ~ 0.1 per cell.

▲ Similarly, Hall measurement shows fewer mobile carriers

- SrTiO_3 is a bulk SC, but can only support few carriers.
Matching $\Rightarrow$ can get interface SC up to 16 layers!

- Consider $\text{LaAlO}_3/\text{SrTiO}_3$



$$E_c = n\Delta + \frac{1}{2} E_{\text{comp}} n^2$$

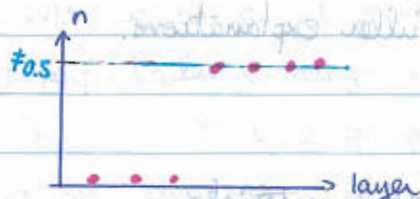
$\sim e^2/\epsilon$

At the other surface we need surface charge n
surface dipole cancel Δn

Combining, $E = \underbrace{n\Delta}_{\text{cost for } e^- \text{ transfer}} + \underbrace{\frac{1}{2} E_{\text{comp}} n^2}_{\text{interlayer repulsion}} - \underbrace{NE_p(1-2n)n}_{\text{energy from } \Delta V} + \underbrace{\frac{1}{2} E_D [N(1-2n)]^2}_{\text{dipolar energy cost at other end of interface}}$

$$\Rightarrow \frac{n}{\text{cell}} = \frac{(NE_p + 2N^2 E_D - \Delta)}{(E_{\text{comp}} + 4NE_p + 4N^2 E_D)}$$

- In experiment



Classically, $E = \Delta \sum_j n_j + \frac{1}{2} E_c \sum_j n_j^2 + \frac{2\pi e^2}{\epsilon c} \sum_{j=0}^{\infty} \sum_{l=j+1}^{\infty} n_j n_l |j-l|$

Putting all n in 0^{th} layer, $E = n\Delta + \frac{1}{2} E_c n^2$

If we have $n-\delta$ in layer 0, & δ in layer 1

$$\Rightarrow \Delta E = \delta(E_p - E_c)n + O(\delta^2)$$

Thus only delocalized charge if $E_c > E_p$ [hard!]

▲ We've neglected e^- hopping, which will produce the desired charge spreading.